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## XRD of Polymers

### Objective:

The objective of this lab is to become familiar with the use of x-ray diffraction in polymer analysis. The emphasis is on the difference between XRD in polymers and XRD in metals and ceramics. The main uses in polymers are:

- 1) determination of unit cell type and lattice parameters
- 2) determination of the degree of crystallinity (DOC)
- 3) determination of the microstructure through the Scherrer equation
- 4) determination of crystallographic orientation through pole figures and the Hermans orientation function

The department does not own a transmission diffractometer which is the most useful for polymer analysis. Several Statton photographic cameras as well as a reflection geometry spectrometer (designed for metals and ceramics) are available. We also have access to Image plate based diffractometers at Wright Labs and Procter & Gamble whose data will be used in this lab for construction of a pole figure and calculation of the Hermans orientation function.

**Note: The prerequisites for this course require some knowledge of XRD equivalent to an undergraduate course and lab in XRD, a course on analytic techniques as well as an introductory course on polymers. Please see texts by Cullity, Alexander, Vonk, and Schultz for descriptions of XRD in polymeric systems.**

### Instruments to be used:

Phillips Diffractometer (Departmental)  
Departmental Statton Cameras with Polaroid Film (Departmental)  
Image plate diffractometer (P&G and Wright Labs by special arrangement)

### Materials:

Blown films of HDPE, LDPE, LLDPE, Exxon Exceed Resin (Metallocene) and some blends. You can add other polymers studied in this course such as polyhydroxybutyrate or PMMA samples.

### Procedure:

- 1) Take photographic diffraction patterns on the Statton camera for all samples noting the machine direction for blown films on the photograph.
- 2) Use the Statton photographs as a guide for several diffractometer scans for each sample.
- 3) Obtain diffractometer scans for samples which have been stretched to different extents.
- 3) Obtain image plate, azimuthal averaged and radial averaged data from several oriented polymer samples.

### Analysis:

- 1) Determine unit cell type and identify all reflections in the samples studied.
- 2) Calculate lattice parameters for all samples
- 3) Construct a plot of unit cell dimensions, a, b, c, as a function of elongation for stretched films.
- 4) Determine the degree of crystallinity (DOC) for all samples.
- 5) From the diffractometer traces and the image plate data determine the lamellar thickness in the plane normal direction using the Scherrer equation
- 6) From these values calculate the lamellar thickness and approximate the melting point using the Gibbs-Thompson equation (Hoffman-Lauritzen equation).
- 7) From the image plate radial average data calculate the Hermans orientation function for several reflections.
- 8) Using these values calculate the orientation function for the unit cell directions and demonstrate that for orthogonal unit cells the sum of these orientation functions is close to 0.
- 9) Construct a pole figure for one reflection using the image plate data.

### Questions:

- 1) How do your lattice parameters compare with literature values?

- 2) How do you think the degree of crystallinity measured in diffraction would compare with that measured in a density gradient column?
- 3) Explain why the breadth of a diffraction peak might be related to the thickness of lamellae?
- 4) How could micron scale grained inorganics such as silica be identified in a commercial PE sample by diffraction? Is there any evidence for inorganics in the PE samples you studied?
- 5) Describe the qualitative difference between an oriented and an unoriented 2-d diffraction pattern.
- 6) Why are pole figures used for processed polymer samples while Wulff nets are used for single crystal samples such as a silicon boull (sp?)?
- 7) Explain why the Hermans orientation function and some unit cell parameters change with strain.
- 8) Why do some unit cell parameters change dramatically with strain while others remain fairly unchanged?